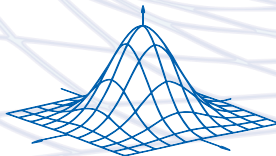


On-the-fly, off-lattice KMC simulations on experimental time scales with k-ART

Peter Brommer

Molecular and Materials Modelling Summer School
27 September 2016

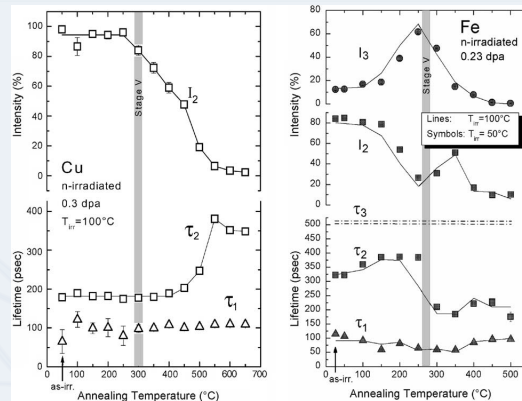
THE UNIVERSITY OF
WARWICK



Point defect complexes

Motivation

- Irradiation causes defect cascades.
- Leaves behind point defects:
 - self-interstitial atoms (SIA)
 - vacancies
- and complexes:
 - dislocation loops
 - stacking fault tetrahedra
 - nanovoids
 - ...



Eldrup and Singh, *J. Nucl. Mater.* **323**, 346 (2003)

- Wealth of defect clusters and events: impossible to predict.
- Time scale is beyond MD (milliseconds – hours).
- Complex energy landscape.

1 Kinetic Activation Relaxation Technique

- Kinetic Monte Carlo
- off-lattice
- self-learning
- handles small barriers

2 Applications

- Vacancies in α -iron
- Amorphous silicon
- Annealing in ion-damaged silicon
- Vacancies near Cu–Zr interface

The kinetic Activation-Relaxation Technique

Kinetic Monte Carlo method

Execute events according to KMC rules.

off-lattice

- Not constrained to lattice (more systems).
- Account for long-range elastic effects.

self-learning

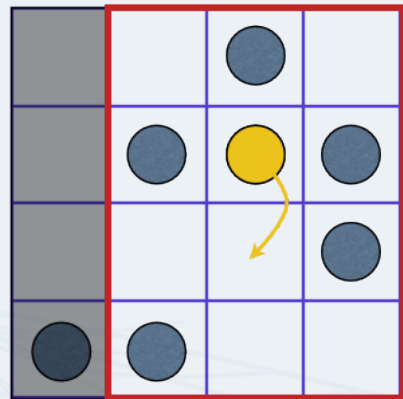
- ART nouveau (fast unbiased saddle point search) to generate events
 - on the fly
 - corrected for long-range effects.
- Store events: Build topology-based catalog.

El-Mellouhi, *PRB* **78**, 153202 (2008). Béland *PRE* **84**, 046704 (2011).



Standard KMC

- Problem must be lattice based.
- List of possible events is constructed
- Rate r_i from transition state theory:
$$r_i = r_0 \exp(-\Delta E/k_B T).$$
- One event picked at random.
- Clock advanced by $\Delta t = -\ln \mu / \sum_i r_i$,
 μ : Random number $\in (0; 1]$.



A.B. Bortz, M.H. Kalos, J.L. Lebowitz, J. Comput. Phys. (1975).

Limitations

- Predefined, **limited** catalogue of known events at $T = 0$.
- Ignores **long-range** interactions between defects.

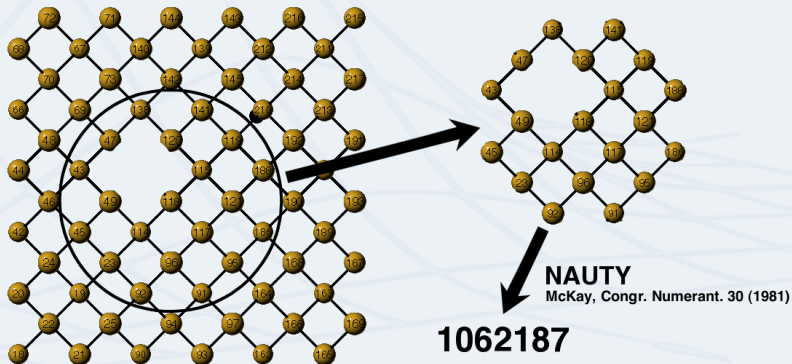
Topologies

Cluster centered on each atom

- Topological analysis: Which atoms are neighbours?
- Assign a key to each graph.

⇒ 1:1 relationship between keys and local structures.

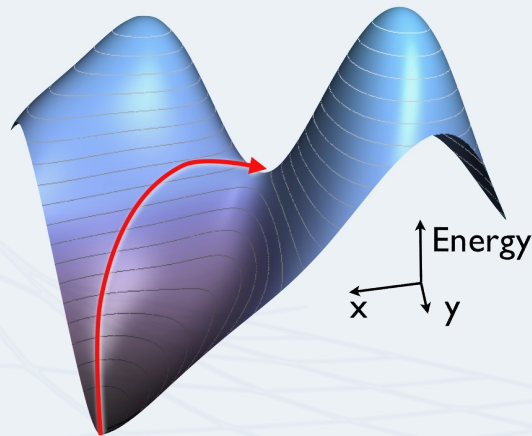
Search for events for each topology.



Find saddle points with ART nouveau

Activation-relaxation technique

- 1 Random displacement.
- 2 Leave harmonic well: negative eigenvalue.
- 3 Push up along corresponding eigendirection, minimize energy in perpendicular hyperplane.
- 4 Converge to saddle point.
- 5 Move configuration over the saddle point and relax to new minimum.



Barkema, Mousseau, PRL 77 (1996); Malek, Mousseau, PRE 62 (2000);

Search for events

Find events centered on representative atom.

- Random displacement.
- Find saddle point (Lanczos, DIIS).

Expensive, but finds **generic** events for topology.

For lowest 99.99% of barrier weight:

Refine event for each **specific** atom.

- Few iterations to exact critical points.
- Takes into account specific local situation.

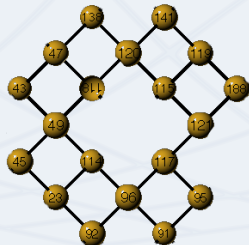
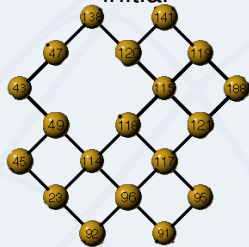
Tree of events

- Calculate rates $r_i = r_0 \exp(-\Delta E_i/k_B T)$, $r_0 = 10^{13} \text{ s}^{-1}$.
- Use tree to select event with proper probability.

Reconstructing events

Geometric transformation

Stored event
initial

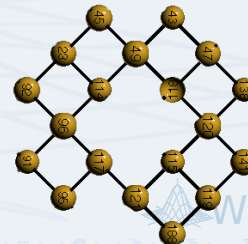
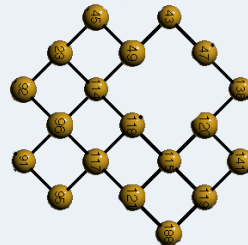


- Extract **symmetry operation** needed to transform stored event to configuration.



- Apply same operation to final (saddle) state.

Configuration



Remembering events

Generic events

- Kept, even though the topology might disappear, but removed from tree.
- Topology reappears: Events reinserted to tree.
- Generic events can be imported from previous runs.

Atom keeps topology

Specific events:

- refined.

Atom changes topology

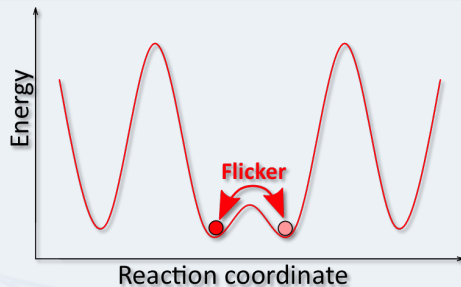
Specific events:

- Old ones removed.
- New ones calculated.

Béland, Brommer, *et al.*, *Phys. Rev. E* **84**, 046704 (2011).

Local configurations with low barriers

- k-ART might get trapped.
- Many events, no progress.



Requirements

- Correct distribution of exit states.
 - Low overhead.
- ⇒ The basin auto-constructing Mean Rate Method
MRM: Puchala *et al.*, *J. Chem. Phys.* **132**, 134104 (2010)

The Mean Rate Method (MRM)

Transient states



Absorbing states

connected by low barriers.

connected to transient states by high barriers.

Basin exploration

- costly
 - even unnecessary (early exit to absorbing state)
- ⇒ Explore/construct basins **on the fly!**

Relevant entities: events, not states

basin event



exit event

connects transient states.

connects transient state to absorbing state.



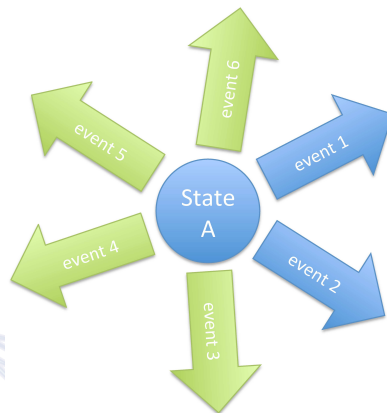
The Basin Mean Rate Method

Start from State A

- Identify events.
- If any event could be a basin event (judge by barrier):
activate basin method.

Pick an event:

- **Ordinary event**: Go on normally
- **Potential basin event**: Start basin:
 - Execute event
 - Block event
 - Keep all other events.



Legend

Green: Ordinary event

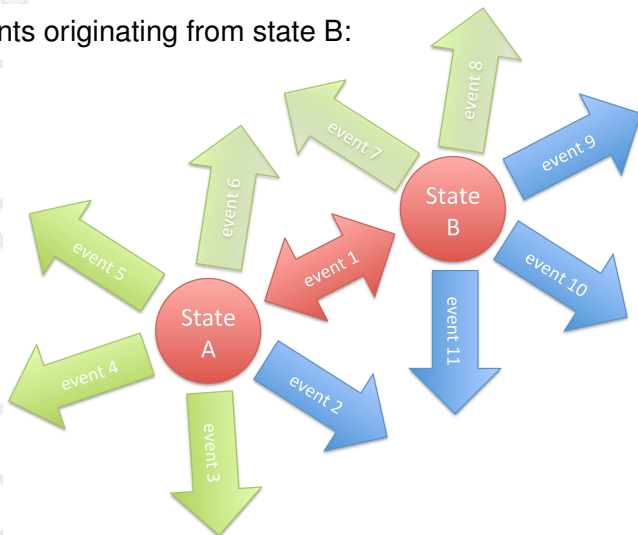
Blue: Potential basin event

Red: Basin event



In the basin

Search for new events originating from state B:



Legend

Green: Ordinary

Blue: Potential basin

Red: Basin

The basin auto-constructing Mean Rate Method

Features

- Basin is built **on the fly**.
 - Basin explored only as far as needed.
 - Integrates seamlessly into k-ART.
- No state is visited twice.
- Averages over “low” barriers.
- Correct distribution of absorbing states.
- However: Ignores correlation between basin residence time and absorbing state (short residence time: absorbing state closer to initial state).

1 Kinetic Activation Relaxation Technique

- Kinetic Monte Carlo
- off-lattice
- self-learning
- handles small barriers

2 Applications

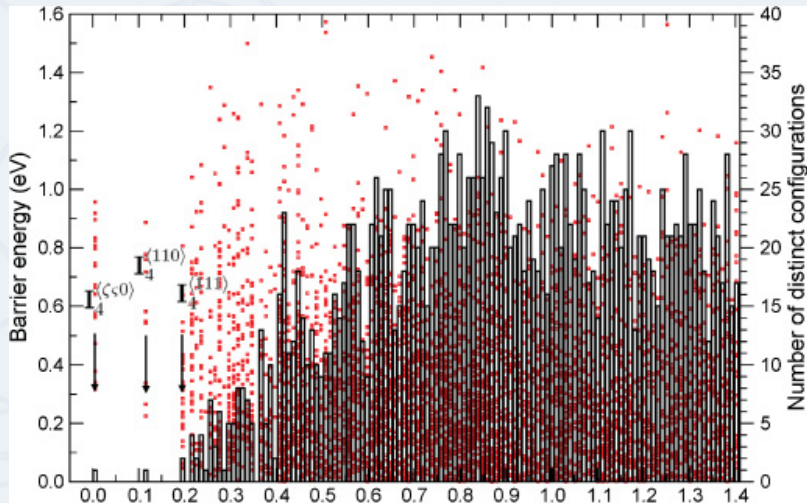
- Vacancies in α -iron
- Amorphous silicon
- Annealing in ion-damaged silicon
- Vacancies near Cu–Zr interface

Atomistic simulation of α -Fe: Challenges

Kinetic Monte Carlo simulations of α -Fe

Extremely rich in states and events:

e.g. 4-SIA cluster: more than 1500 distinct configurations.



Vacancies

Vacancy cluster agglomeration in bcc Fe

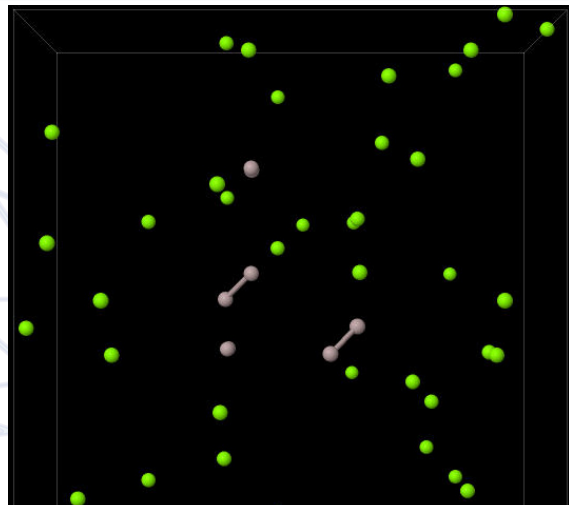
- Slower dynamics than interstitials.
- PAS results available.

The system: 2000 atoms

Remove atoms at random.

- Temperature 50°C.
- Display only vacancies, color code cluster size, green: monovacancies.
- Ackland-Mendelev potential (optimized).

Ackland *JP:CM* **16**, S2629 (2004)



50 vacancy k-ART simulations at 50 °C

Cluster growth

Average size > 6

• 0.1 ms

Time scale

Vacancy clustered:

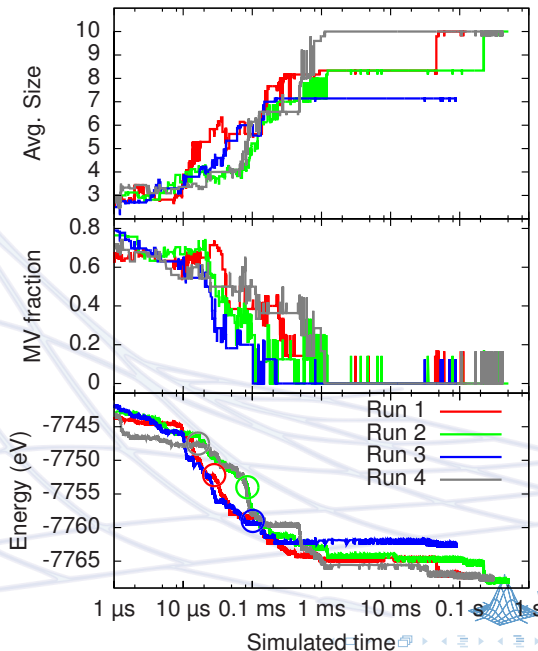
• 1 ms

Energy barriers

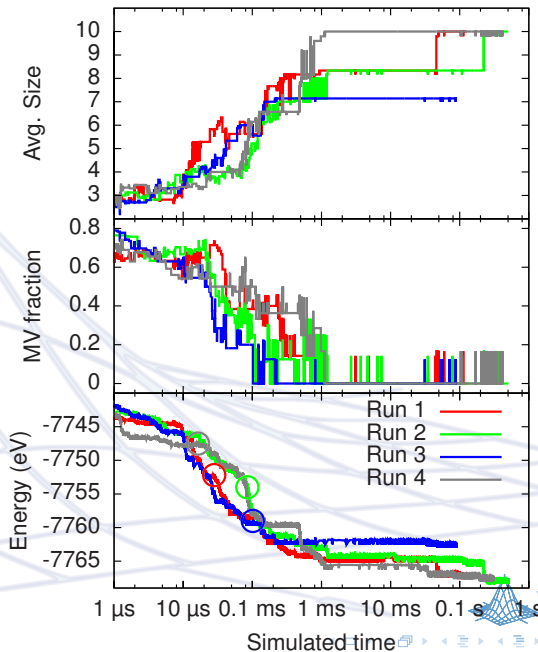
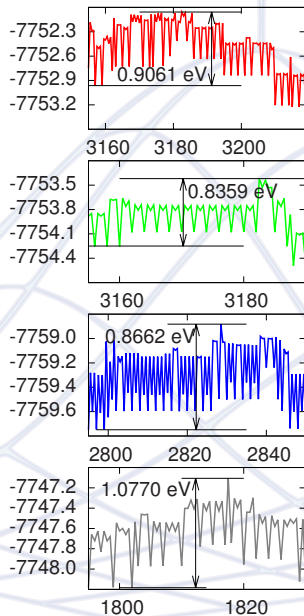
Maximal eff. barrier:

• 0.8–1.1 eV

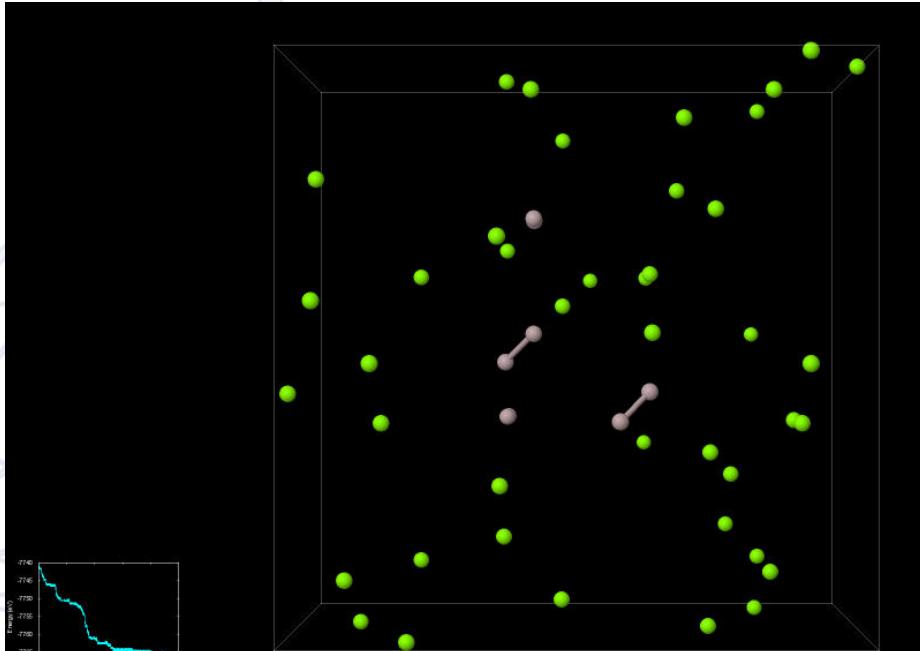
Brommer & Mousseau, PRL **108**,
219601 (2012)



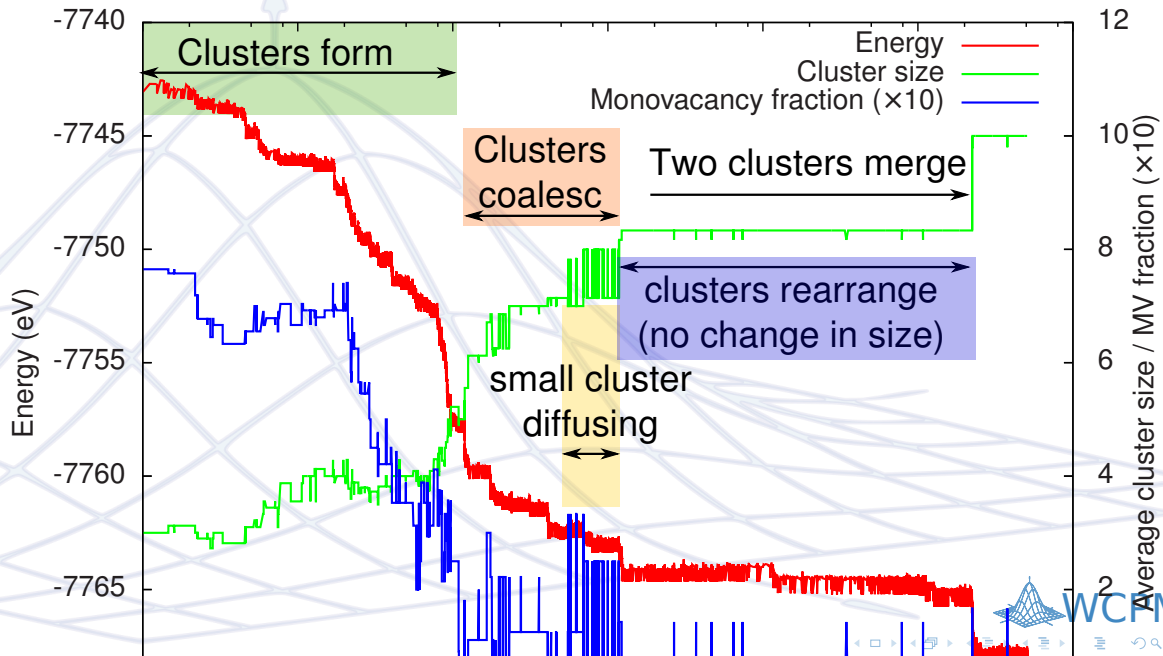
50 vacancy k-ART simulations at 50 °C



50 vacancies in α Fe



Trajectory in detail



Positron Annihilation Spectroscopy

Iron irradiated at 50°C:^a

- Significant intensity from **nanovoids** as irradiated (nanovoids: clusters of 9–14 vacancies).
 - Annealing over 150°C: Larger voids appear (40–50 V)
- ⇒ k-ART simulation agrees with experiment

^aEldrup and Singh, *J. Nucl. Mater.* **323**, 346–353, 2003.

Previous results: Autonomous Basin Climbing

- ABC^a always picks lowest new barrier.
- k-ART may pick higher barrier, accounts for multiplicity.

Complete catalog essential for material description.

^aFan *et al.*, PRL **106**, 125501 (2011)



Energy release

More energy in cluster shape than cluster size

Typical cluster binding energy:

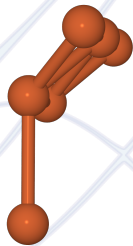
- 0.2 eV

also $\approx$ divacancy binding energy.

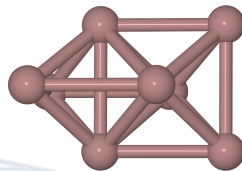
Reshaping a cluster:

- more than 1 eV

Example: di- & hexavacancy



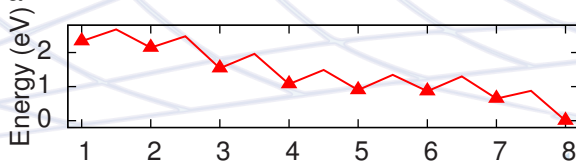
1



8

Initial

Final



Energy release

More energy in cluster shape than cluster size

Typical cluster binding energy:

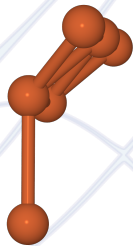
- 0.2 eV

also $\approx$ divacancy binding energy.

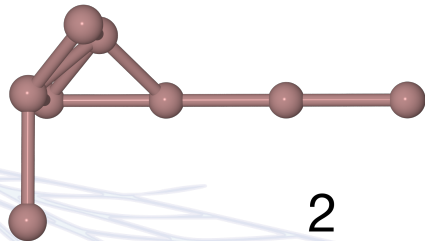
Reshaping a cluster:

- more than 1 eV

Example: di- & hexavacancy

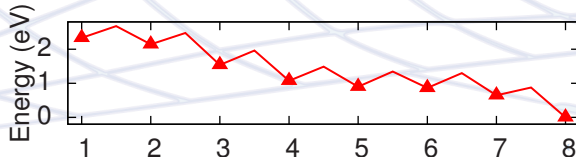


1



2

Simulated time: 4.54 μ s — Barrier: 0.342 eV — ΔE : -0.191 eV



Energy release

More energy in cluster shape than cluster size

Typical cluster binding energy:

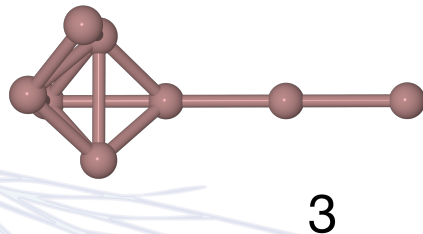
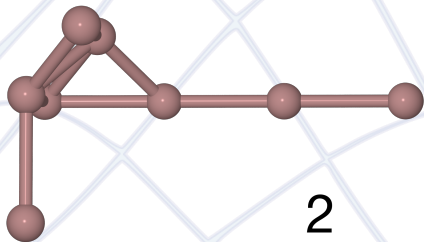
- 0.2 eV

also $\approx$ divacancy binding energy.

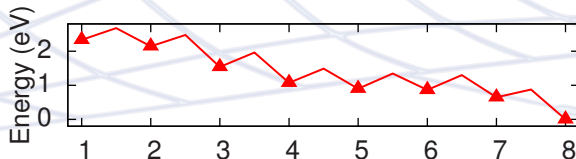
Reshaping a cluster:

- more than 1 eV

Example: di- & hexavacancy



Simulated time: 4.62 μ s — Barrier: 0.334 eV — ΔE : -0.603 eV



Energy release

More energy in cluster shape than cluster size

Typical cluster binding energy:

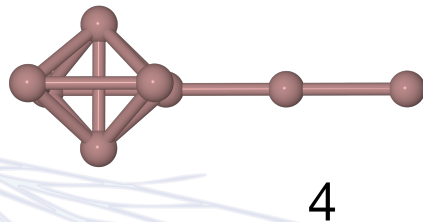
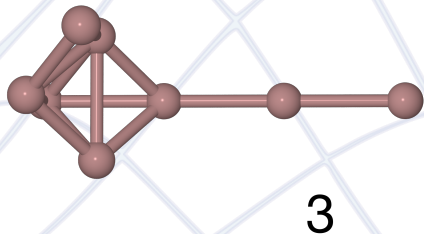
- 0.2 eV

also $\approx$ divacancy binding energy.

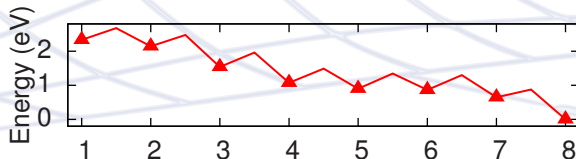
Reshaping a cluster:

- more than 1 eV

Example: di- & hexavacancy



Simulated time: 5.61 μ s — Barrier: 0.421 eV — ΔE : -0.464 eV



Energy release

More energy in cluster shape than cluster size

Typical cluster binding energy:

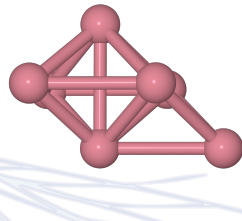
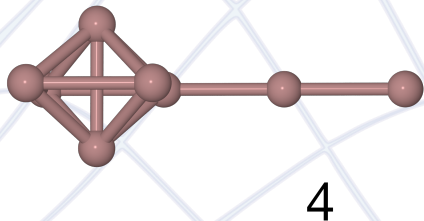
- 0.2 eV

also $\approx$ divacancy binding energy.

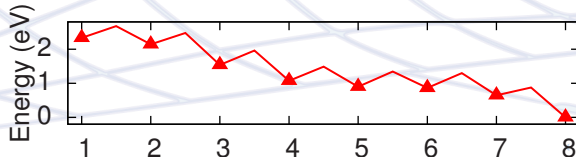
Reshaping a cluster:

- more than 1 eV

Example: di- & hexavacancy



Simulated time: 6.14 μ s — Barrier: 0.414 eV — ΔE : -0.177 eV



Energy release

More energy in cluster shape than cluster size

Typical cluster binding energy:

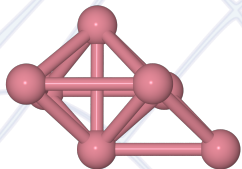
- 0.2 eV

also $\approx$ divacancy binding energy.

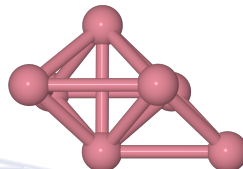
Reshaping a cluster:

- more than 1 eV

Example: di- & hexavacancy



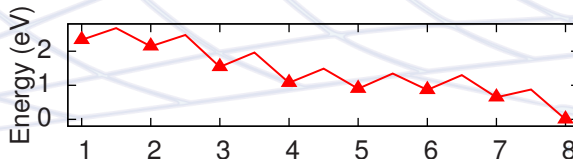
5



6



Simulated time: 6.26 μ s — Barrier: 0.445 eV — ΔE : -0.033 eV



Energy release

More energy in cluster shape than cluster size

Typical cluster binding energy:

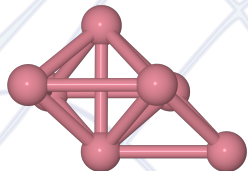
- 0.2 eV

also $\approx$ divacancy binding energy.

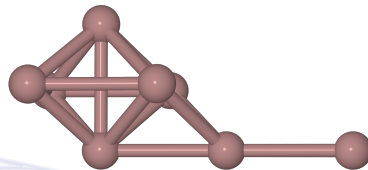
Reshaping a cluster:

- more than 1 eV

Example: di- & hexavacancy

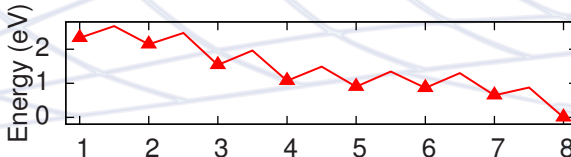


6



7

Simulated time: 7.11 μ s — Barrier: 0.434 eV — ΔE : -0.216 eV



Energy release

More energy in cluster shape than cluster size

Typical cluster binding energy:

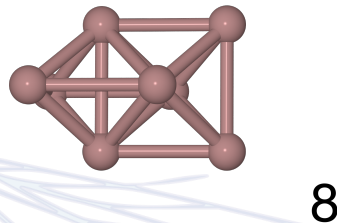
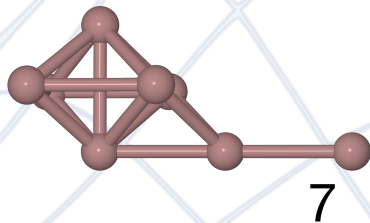
- 0.2 eV

also $\approx$ divacancy binding energy.

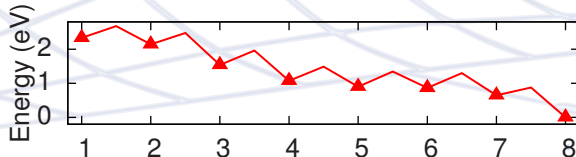
Reshaping a cluster:

- more than 1 eV

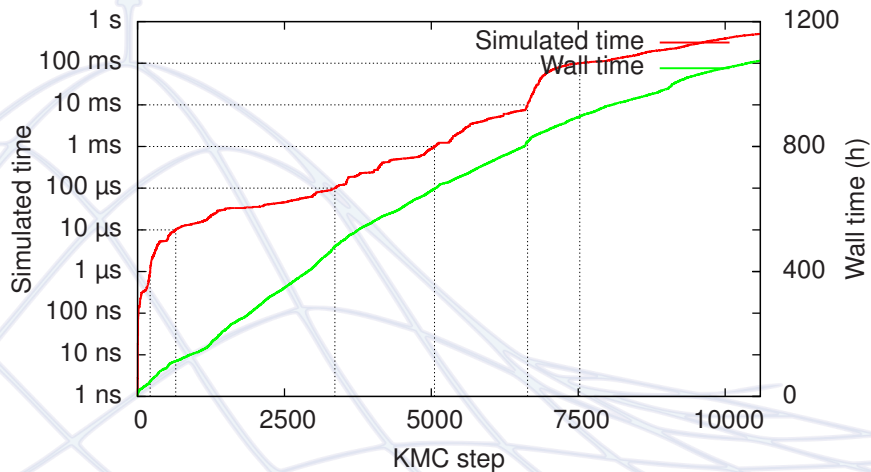
Example: di- & hexavacancy



Simulated time: 17.19 μ s — Barrier: 0.228 eV — ΔE : -0.649 eV



Accelerating simulation



Reasons

- 1 Lower effective energy barriers **die out**.
- 2 Basin **acceleration** threshold increased.

Increasing basin threshold

Optimal basin threshold

There is an optimal value for the basin threshold:

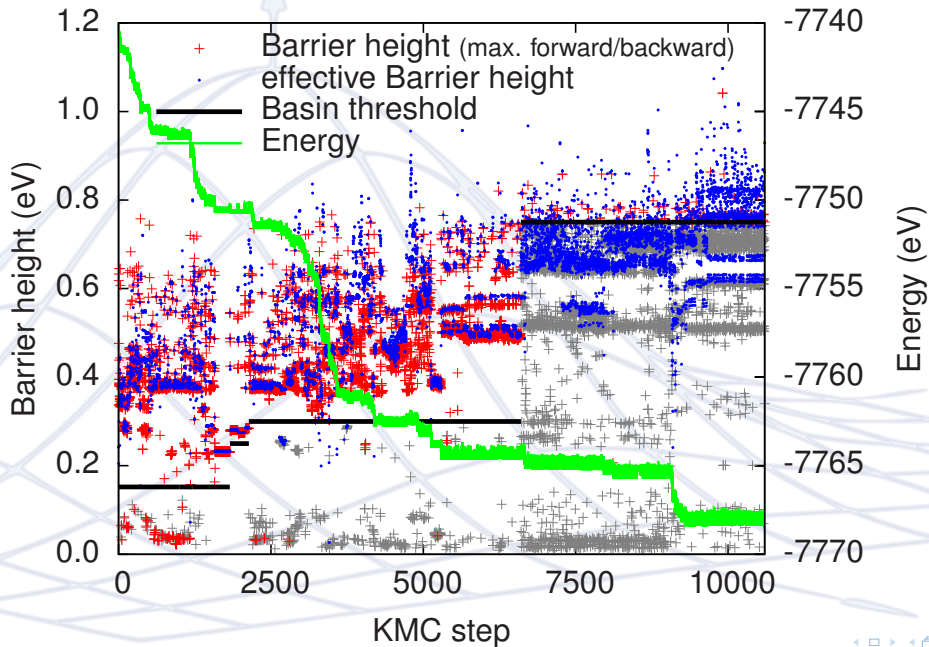
Too low: No progress.

Too high: Too many states in basin:

- Lose trajectory
- Memory requirements.

Gradual increase during simulation.

Basin Threshold



Conclusions: α -iron

Vacancies in bcc iron

- Vacancies cluster in nanovoids on a sub-second timescale.
- Full event catalog essential.
- Efficiently accelerated by bac-MRM. Brommer *et al.*, Phys. Rev. B **90**, 134109 (2014)

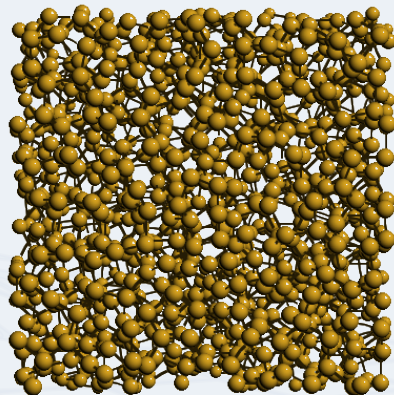
Amorphous silicon

Disordered metastable phase of Si

Defects in amorphous silicon:

- Are vacancies stable defects?
- Do vacancies diffuse?

No accelerated technique has been applied to disordered materials.



Project of Jean-François Joly (Ph.D. student UdeM, then Post-Doc Ottawa/Carleton).

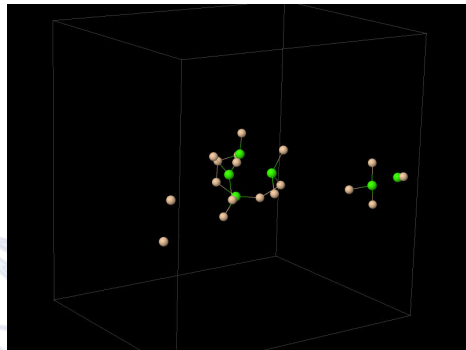
k-ART simulation of a vacancy in a-Si

The system

- 999 (= 1000 – 1) atoms.
- Mod. Stillinger-Weber potential.
- $T = 300$ K

Challenges

- Every atom: unique topology. Initial catalog: 32 120 events.
- Flickers on every energy scale.
Basin threshold: 0.35 eV.

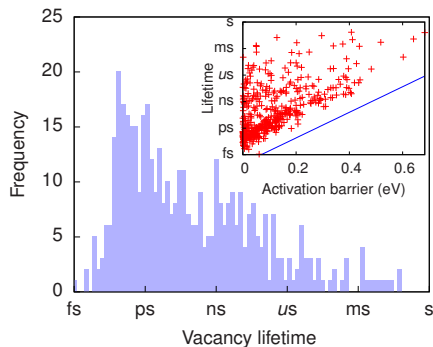


Results: Vacancies in a-Si

1000 k-ART runs

(one for each site as vacancy)

- Most times:
Vacancy disappears
 - in initial minimization (547)
 - or first few steps (ns) (175).
- Survivors
 - 1 μ s: 47
 - 1 ms: 13
 - at the end: 6 for 0.12–1.98 s.



Joly, PB, *et al.*, PRB **87**, 144204 (2013)

Vacancy diffusion

- Vacancy diffusion in 86 runs:
 - Single step, then annihilation: 56
 - Flickers until annihilation: 30

⇒ Vacancy diffusion not relevant, enough other defect modes.

Ion-damaged crystalline silicon

Experiment

Nanocalorimetry on ion-damaged silicon.

- 330 nm thick crystalline silicon strip.
- Implanted with Si^- , 10–100 keV.
- Heated at 10^5 K s^{-1} .
- Measure heat release.

Simulation

MD+kART:

- 3 keV Si atom into 100 000-atom slab of SW Si (300 K, Langevin bath).
- Cut out 27 000 atoms with defects, PBC.
- Sample kART trajectories.
- Calculate heat release at constant T , transform.

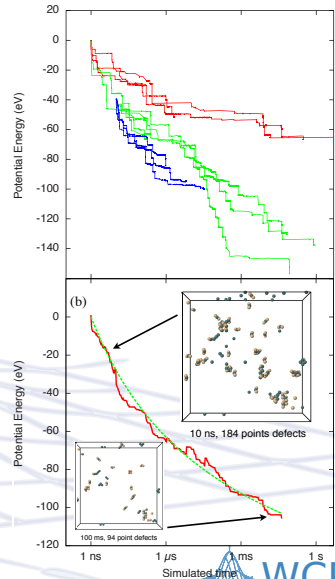


kART simulations

- 3 MD implementation runs
- 3 kART trajectories for each MD.
- Bottom: Average.
- Green: Fit of event distribution.

Replenish and relax:

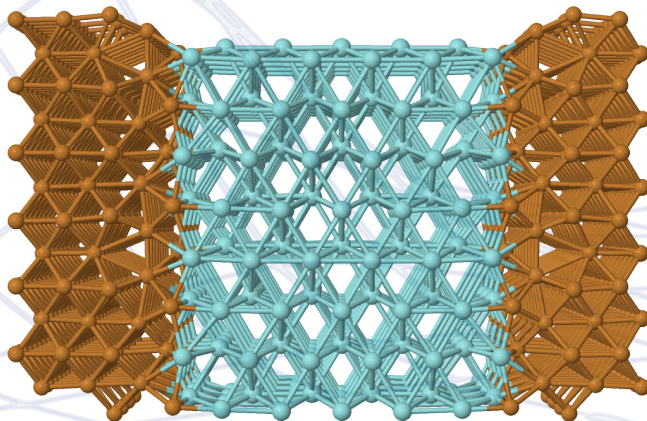
- Logarithmic relaxation.
- High barrier events: Not necessarily large relaxation.
- But: Access to new areas of configuration space, from which further relaxation possible.
- High barriers: Structural rearrangements.



k-ART for alloy systems

Kinetic ART can also be applied alloy systems.

- Example: Vacancy dynamics near Cu–Zr interface.



Cu-Zr system

Test simulations:

- 976 atoms (624 Cu, 352 Zr); 5824 atoms (3648 Cu, 2176 Zr).
- Cu 110 on Zr 11 $\bar{2}$ 0 (minimal lattice mismatch).
- 1 vacancy at different distance from interface.
- EAM effective potential: Cheng *et al.*, PRL 102, 245501 (2009).

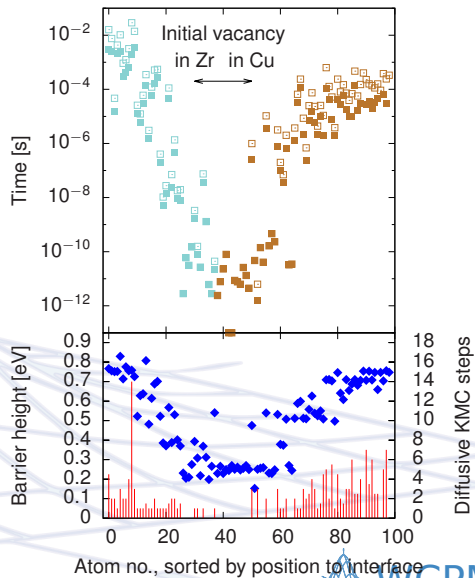
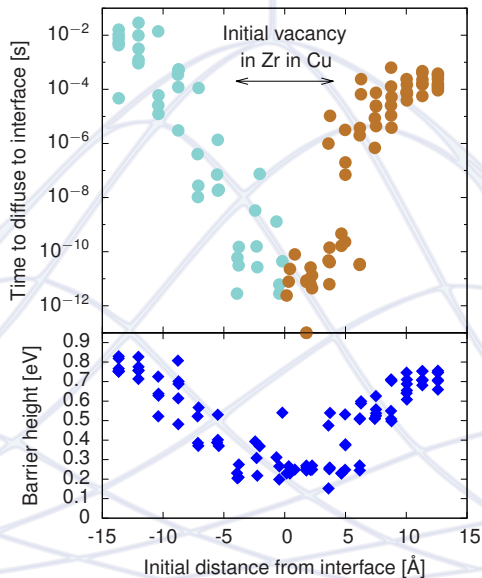
Kirkendall effect

Difference in diffusivity between metals at metal/metal interface

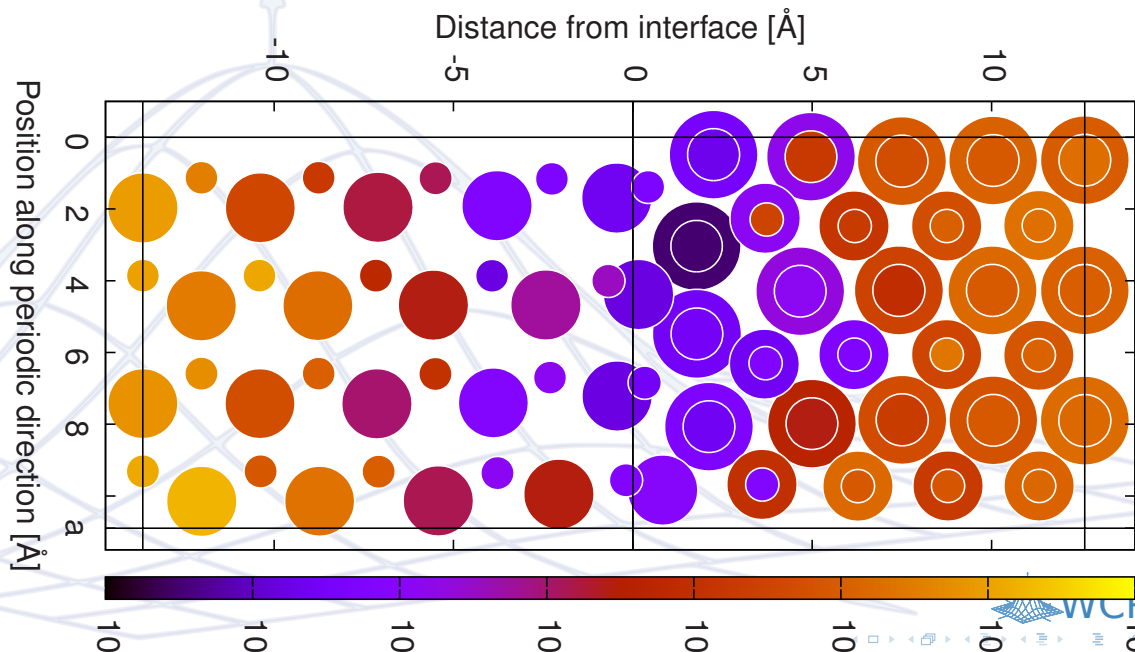
- Interface moves.
- Kirkendall voids at interface.

Test simulations: can k-ART do that?

Diffusion time to interface



Diffusion time from each site



Kinetic Activation-Relaxation Technique (k-ART)

Versatile KMC simulation tool for complex systems:

- Off-lattice, self-learning: Few prerequisites.
- Fully account for long-range elastic effects.
- Can handle feature-rich defect systems.
- Basin treated with bac-MRM.
- Even fully amorphous systems.

El-Mellouhi *et al.*, *Phys. Rev. B* **78**, 153202 (2008).

Béland, Brommer, *et al.*, *Phys. Rev. E* **84**, 046704 (2011).

Acknowledgements

Acknowledgements

- N. Mousseau, L. Lewis, J.-F. Joly, L.K. Béland (U de Montréal)
- F. El-Mellouhi (Texas A&M @ Qatar)
- M.-C. Marinica (CEA Saclay, France)

Funding



Fonds de recherche
Nature et
technologies

Québec



Thank you for your attention!

